



Visible-light photocatalysis with phosphorus-doped titanium(IV) oxide particles prepared using a phosphide compound

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ABSTRACT

Doping titanium(IV) oxide (TiO₂) with phosphorus (P) having negative charges was attempted in order to obtain P-doped TiO₂ with a red shift of the absorption edge and visible-light photocatalytic activity. P-doped TiO₂ was synthesized through the hydrolysis of titanium tetraisopropoxide in a suspension containing phosphide. The P-doped TiO₂ nanoparticles exhibited a pale yellow color. The UV–vis diffuse reflection spectra of P-doped TiO₂, which was calcined at 450–600 °C, showed a continuous and tailing absorption in the visible region. XRD patterns of P-doped TiO₂ showed the presence of anatase phase up to a temperature of 500 °C and the appearance of rutile phase from a temperature over 550 °C. The red shift of the absorption edge was not associated with the phase transformation from anatase to rutile. The XPS P2p spectra of P-doped TiO₂ indicated the decrease of P⁵⁺ and the increase of P^{3–} during sputtering. The results suggested the presence of phosphate on the surface and the presence of P^{3–} on the inner side of P-TiO₂. The action spectra of photocatalytic reaction for acetaldehyde decomposition revealed visible light activity of P-doped TiO₂. P-doped TiO₂ calcined at 475 °C was the most active in degradation of phenol under visible light. The P-content was found to play a key role in determining the spectral response in visible area and the photocatalytic activity under visible light irradiation.

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1. Introduction

Titanium(IV) oxide (TiO₂) has been extensively used and investigated for the remediation of environmental organic pollutants because of the advantage of being a nontoxic, stable and inexpensive photocatalyst [1–3]. However, its wide band gap (3.2 eV for the anatase structure) allows it to absorb only ultraviolet light, which accounts for only ~6% of solar photons. A large number of approaches have been taken to reduce the band-gap energy of TiO₂ by doping with transition metal cations [4,5] and non-metal anions of elements such as N [6–16], C [17,18] and S [19,20]. N-doped TiO₂ has been synthesized by using NH₃ [6–9], urea [10–12], ethylenediamine [13], ethylmethanamine [14] or NH₄Cl [15,16] as a precursor of N. The properties of N-doped TiO₂ have already been discussed with respect to the nitrogen content [8], oxidative power [9], desorption of nitrogen from the O–Ti–O matrix in annealing [14], characterization of different prepared materials [15], increase of carrier-recombination centers in TiO₂ [11], and the effect of Fe₂O₃ nanoparticles on N-doped TiO₂ [12].

In general, phosphorous doped (P-doped) TiO₂ prepared using phosphate is not able to absorb visible light. Some groups [21–24] have reported that P-doped TiO₂ prepared using H₃PO₄ had a slight larger band gap than that of pure TiO₂. On the other hand, results of several experiments on P-doped TiO₂ prepared using phosphate with band gap narrowing (a red shift of the absorption edge) have been reported [25–28]. In those studies, P-doped TiO₂ was synthesized by a sol–gel method using H₃PO₂(P⁺) [25–27] or H₃PO₄(P⁵⁺) [28] as a precursor of P. The band-gap energies of P-doped TiO₂ prepared using H₃PO₂ and H₃PO₄ were 3.13 eV (absorption edge: 396 nm) [26] and 3.11 eV (399 nm) [28], respectively. Those studies suggested formation of Ti–O–P bonds in the anatase lattice [27,28]. The P species in the titania were considered to be the dominant group responsive to the small change in band gap [27]. Yang et al. reported first-principles calculation for substitutional P-doped anatase and rutile TiO₂ [29]: in the case of doping anatase TiO₂ with phosphorus having positive charges and in the case of doping rutile TiO₂ with phosphorus, the band gaps of the doped TiO₂ had little narrowing. On the other hand, in the case of doping anatase TiO₂ with phosphorus having negative charges, the band gap of the doped TiO₂ had a slight narrowing because some P3p states were considered to exist in it. That study suggested that doping anatase TiO₂ with phosphorus having negative charges made it

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possible to prepare P-doped TiO_2 with visible-light photocatalytic activity.

We have reported the characterization of P-doped TiO_2 prepared by calcination after grinding phosphide (P^{3-}) and TiO_2 materials in a mortar [30]. However, a shift of the absorption edge was not apparent. A possible interpretation is that P species could not be sufficiently introduced into the TiO_2 lattice because phosphide and TiO_2 materials were mixed in solid form. The purpose of the present study was the first synthesis of P-doped TiO_2 with a red shift of the absorption edge and visible-light photocatalytic activity using a phosphide compound as a precursor of P. The mixing process of phosphide and TiO_2 materials was performed in solution. Visible-light activity was investigated using action spectra. The features of visible-light photocatalysis for P-doped TiO_2 are discussed.

2. Experimental

2.1. P-doped TiO_2 preparation

All chemicals used in this study except for a few obtained from companies shown in parentheses were purchased from Wako Pure Chemical Industries, Ltd.

P- TiO_2 was synthesized by calcination of the precipitates after hydrolysis of titanium tetraisopropoxide in a suspension containing phosphide. First, phosphide was synthesized as follows: Titanium tetrachloride was mixed with 2 equiv. of cyclohexylphosphine (Strem Chemicals Inc.) in hexane at ambient temperature. A pure yellow solid of $[\text{TiCl}_4(\text{C}_6\text{H}_{11}\text{PH}_2)_2]$ was separated by filtration [31]. After the yellow solid had been heated in an ampoule under nitrogen atmosphere at 450°C for 6 h, a black solid (denoted as *TiP compound*) was obtained. Then the phosphide and TiO_2 materials were mixed as follows: 0.017 g of the *TiP compound* was dispersed in a mixture of 2.84 g titanium tetraisopropoxide and 6 mL 2-propanol. The following hydrolysis process was achieved by adding 30 mL of deionized water (Milli-Q water, Millipore) dropwise into the solution under stirring. A gray precipitate was produced instantly, and the mixture was stirred for 10 min. The precipitate was separated by filtration and washed 3 times with deionized water. The collected solids were dried at 110°C for 3 h and calcined for 6 h at 450 – 600°C . (These products are denoted as P- TiO_2 (T), T: each calcination temperature.) P_{high}- TiO_2 was prepared in the same way with 10 times higher ratio of P. ST01 (Ishihara Sangyo Kaisha, Ltd.) was used as a reference. PO_4 -adsorbed TiO_2 was prepared as follows: ST01 powder (1.0 g) was added to 100 mL aqueous H_3PO_4 solution (20 mM), and the solution was kept stirring for 1 h in dark. The PO_4 -adsorbed TiO_2 powder was obtained after being filtered and dried at 110°C for 3 h.

2.2. Characterization

The crystal structure of TiO_2 was examined by an X-ray diffraction method (XRD) using Rigaku Multiflex 2 kW with a Cu-K α source. The chemical composition on the surface of TiO_2 was examined by X-ray photoelectron spectroscopy (XPS) using PHI model for 1800 equipped with a monochromatic Al-K α source at a pass energy of 23.5 eV. The core level binding energies were aligned with respect to the C1s binding energy of 285 eV. An electron flood gun was used to compensate for the build-up of positive charge on the insulated samples during the measurement. Ar⁺ sputtering (2 kV) was applied to examine the surface of the samples. UV–vis absorption spectra were recorded by a JASCO V-570 UV–vis spectrometer. The specific surface area of TiO_2 was analyzed by the BET method from N_2 adsorption isotherms at 77 K using BELSORP-max.

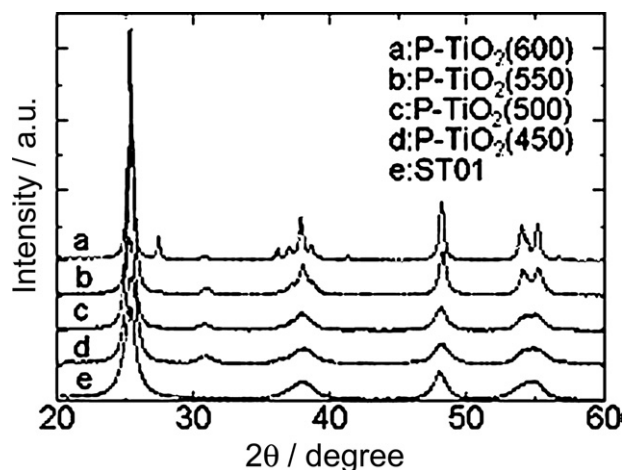


Fig. 1. XRD patterns of ST01 and P- TiO_2 at different calcination temperatures.

The samples were dried at 110°C for 3 h before the adsorption measurement. IR spectra were investigated using JASCO FT/IR-480Plus.

2.3. Action spectra

Action spectra were measured with monochromatic light irradiation in the wavelength range from 350 nm to 470 nm using a diffraction grating-type illuminator (JASCO, CRM-FD) equipped with a 300-W Xenon lamp (Hamamatsu Photonics, C2578-02). The light intensity at each wavelength was detected by a power meter (HIOKI 3664). Full-width at half-maximum intensity (FWHM) of monochromatic light was ca. 15 nm. A container containing the sample photocatalyst (10 mg) and acetaldehyde was sealed and kept in the dark for 2 h to reach an adsorption equilibrium. The molar amounts of CO_2 as a result of photocatalytic decomposition of acetaldehyde were measured by a gas chromatograph (Shimadzu, GC-14B) equipped with an FID and a methanizer (Shimadzu, MTN-1).

2.4. Photocatalytic activity measurements

The photocatalytic activity of P- TiO_2 was determined by measuring the degradation of phenol and the appearance of 1,4-benzoquinone under irradiation. Phenol is one of the model compounds for photocatalytic degradation [32]. An aqueous suspension (5.0 mL) of TiO_2 (8.0 g/L) and phenol (0.050 mM) was prepared. Irradiation was performed with a 450-W incandescent lamp (Iwasaki Electric Co. Ltd.). Wavelengths were selected with a Y-45 glass filter (AGC Techno Glass Co., Ltd.) and an aqueous nickel sulfate solution to emit in the 423–600 nm range. The suspensions were stirred for 15 min prior to and during the irradiation. After irradiation, samples were extracted through a $0.45\ \mu\text{m}$ Sartorius filter (MINISART RC4) for analysis by HPLC (Shiseido Fine Chemicals, Nanospace system). Compounds were separated on a reversed-phase column (Shiseido Fine Chemicals, C18AQ) with H_3PO_4 buffer (0.1 wt.%)/CH₃OH (90:10). Photodiode array detection at 260 nm was used for quantification.

3. Results and discussion

3.1. Photocatalyst structure

First-principles calculation [29] predicted that crystal structure was an important factor for the preparation of P- TiO_2 with visible light photocatalytic activity. Fig. 1 shows XRD patterns of P- TiO_2 and ST01. The XRD patterns of P- TiO_2 up to a temperature of 500°C

Table 1
Crystallite sizes and surface areas of P-TiO₂ and ST01.

Sample	Crystallite size (nm)	Surface area (m ² g ⁻¹)
ST01	8.8	297
P-TiO ₂ (450)	8.1	127
P-TiO ₂ (500)	10	99
P-TiO ₂ (550)	18	56
P-TiO ₂ (600)	37	29

correspond to the pattern of ST01 which is in the anatase form. Rutile growth begins to appear at 550 °C. Small peaks around 31° indicate the existence of a small quantity of brookite crystal structure. The data are consistent with the results of a study [33] in which sol-gel synthesis of titania produced a mixture of brookite and anatase. Crystallite size was determined using Scherrer's equation [34]:

$$D = \frac{0.89\lambda}{\beta \cos \theta} \quad (1)$$

where λ (nm) is the X-ray wavelength, θ is the Bragg angle, and β is the corrected full width at half maximum (FWHM) of the most intense diffraction peak. The crystallite sizes for P-TiO₂ and ST01 are summarized in Table 1. The size of ST01 was 8.8 nm, comparable to the reported data [35]. The size for P-TiO₂ increased with increase in calcination temperature. The specific surface areas of P-TiO₂ and ST01 are also summarized in Table 1. The surface area of ST01 is in good agreement with the reported data [35]. The surface area of P-TiO₂ decreased with increasing calcination temperature.

3.2. UV-vis absorption spectra

Fig. 2(A) shows absorption spectra before and after calcination. The absorption in the range of 400–500 nm before calcination is attributed to the color of the *TiP* compound itself. The disappearance of absorption in the same range after calcination indicates the formation of a different state for P species.

Fig. 2(B) shows absorption spectra of P-TiO₂ and ST01. All of the synthesized P-TiO₂ samples exhibit a pale yellow color. The UV-vis spectra of P-TiO₂ show a continuous and tailing absorption in the visible region. The absorption edge around 400 nm for ST01 is assigned to electronic transitions from the valance band to conduction band for the anatase phase. The band-gap energies of P-TiO₂(500) and ST01 are 2.94 and 3.09 eV, respectively, which were determined by the plot of the modified Kubelka–Munk function $(ah\nu)^{1/2}$ versus photon energy assuming indirect transition. The spectra of P-TiO₂(500), which is in the anatase and brookite form, are clearly modified by the onset of a relatively broad absorption band in the visible region. The red shift of P-TiO₂ does not correlate with brookite phase because the band-gap energy of brookite is reported to be 3.14 eV [36]. The absorption spectra and XRD analysis of P-TiO₂ indicate that the red shift of the absorption edge is not associated with the phase transformation from anatase to rutile.

3.3. FTIR spectra

Fig. 3 shows FTIR spectra for P-TiO₂ and ST01. The signals in the range of 700–1250 cm⁻¹ are characteristic of formation of the O–Ti–O lattice [15]. The infrared spectra of all samples show signals at 1630 cm⁻¹, 2340 cm⁻¹ and 3400 cm⁻¹ (broad), which are ascribed to OH [15], adsorbed CO₂ [16] and residual adsorbed water [16], respectively. The 1630 cm⁻¹ and 3400 cm⁻¹ peaks decrease with increasing calcination temperatures. A 1300 cm⁻¹ signal, which is characteristic of C–H [14], was not detected. Zhao et al. reported that the signals for organic groups of the raw material

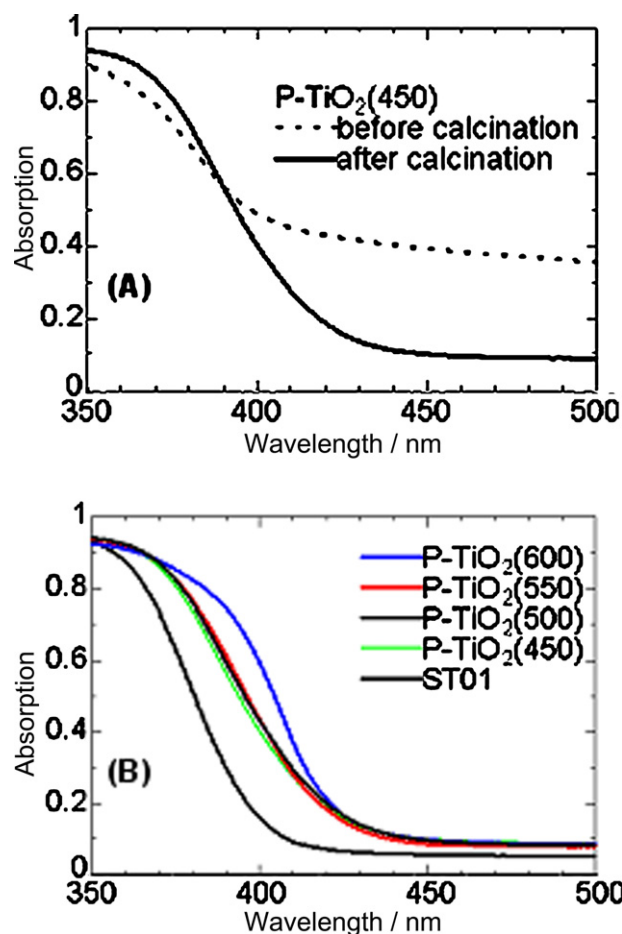


Fig. 2. Absorption spectra of P-TiO₂(450) before calcination and after calcination (A) and absorption spectra of ST01 and P-TiO₂ at different calcination temperatures (B).

in IR spectra disappeared with heating at 400 °C for 30 min in their process for synthesis of TiO₂ [13]. Since P-TiO₂ was calcined at a temperature above 400 °C for 6 h, no peak corresponding to hydrocarbon can be seen in Fig. 3. A 1030 cm⁻¹ peak, which corresponds to PO₄ [37], was not observed in our calcination process despite partial oxidation of P on the surface (see Fig. 4). This suggests that a small quantity of P atoms exists on the surface of TiO₂ (see Table 2).

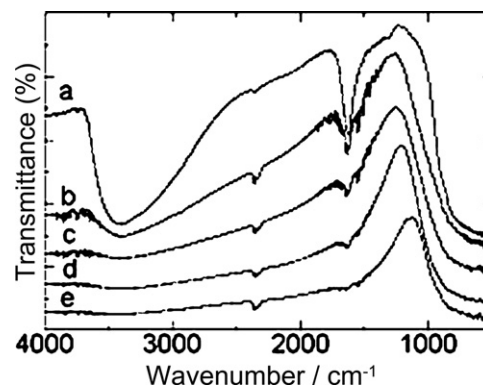


Fig. 3. IR spectra of ST01 (a), P-TiO₂(450) (b), P-TiO₂(500) (c), P-TiO₂(550) (d), and P-TiO₂(600) (e).

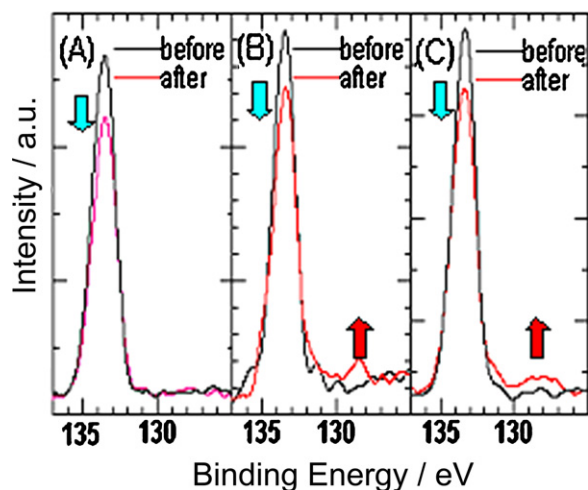


Fig. 4. XPS P2p spectra of PO₄-adsorbed TiO₂ (A), P-TiO₂(500) (B), and P_{high}-TiO₂ (C) before and after sputtering.

3.4. XPS analysis

Fig. 4 shows P2p spectra of P-TiO₂, P_{high}-TiO₂, and PO₄-adsorbed TiO₂ before and after sputtering. A peak at 134 eV is assigned to P⁵⁺ [22]. The decrease of the peak intensity for PO₄-adsorbed TiO₂ during sputtering in Fig. 4(A) indicates the presence of phosphate on the surface of TiO₂. The same behaviors of the peaks at 134 eV were observed for P-TiO₂ and P_{high}-TiO₂ in Fig. 4(B and C). These results indicate the presence of phosphate on the surface of P-TiO₂ and P_{high}-TiO₂ resulted from the partial oxidation of phosphide during our synthetic process. A peak at approximately 128 eV is assigned to P³⁻ [22]. The increase of the peak intensities at 128 eV for P-TiO₂ and P_{high}-TiO₂ during sputtering indicates the presence of P³⁻ on the inner side of TiO₂. Furthermore, the color change from gray to pale yellow during calcination as shown in Fig. 2(A) indicates the formation of different states for phosphorus atoms because gray is attributed to the TiP compound of raw materials. Theoretical calculation [29] predicted that visible light absorption of P-TiO₂ required the replacement of O atoms by P atoms having negative charges in the TiO₂ anatase matrix. Thus, the detection of P³⁻ in P-TiO₂, color change and theoretical calculation suggest that the dopant P is located at a substitutional site for an O atom in the anatase TiO₂ crystal. The red shift of the absorption edge of P-TiO₂ is interpreted as band gap narrowing induced by P³⁻ atoms in the TiO₂ anatase matrix.

The surface composition of P-TiO₂ was calculated using the sensitivity coefficient from peak areas of the XPS analysis. The composition is summarized in Table 2. The value of P/Ti for P-TiO₂(600) is larger than those for P-TiO₂(450, 500, 550). Nitrogen-content of nitrogen-doped TiO₂ usually decreased with the enhanced calcination temperature because of desorption of nitrogen from the O–Ti–O matrix in annealing [13]. The behavior of P-content on the surface of P-TiO₂ was found to be different from that of

Table 2
Surface composition of P-TiO₂.

Sample	Ti (%)	P (%)	P/Ti ^a (XPS)	P/Ti ^b (wash)
P-TiO ₂ (450)	30.1	0.54	0.018	0.018
P-TiO ₂ (500)	29.4	0.53	0.018	0.019
P-TiO ₂ (550)	30.5	0.55	0.018	0.020
P-TiO ₂ (600)	30.6	1.02	0.033	0.035

^a The values were calculated using XPS data.

^b The values were obtained as follows: an aqueous suspension (100 mL) containing P-TiO₂(50 mg) was stirred for 1 h. Next, the TiO₂ particles were separated by filtration. Then the values of washed P-TiO₂ were calculated using XPS data.

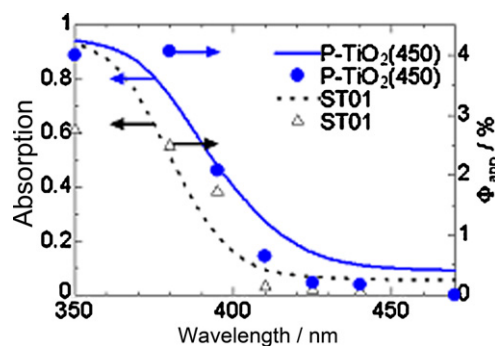
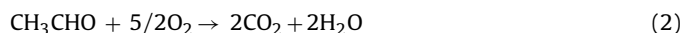


Fig. 5. Absorption and action spectra of P-TiO₂(450) (bold line and •) and ST01 (dotted line and Δ).

N-content on the surface of N-doped TiO₂. The values of P/Ti in P-TiO₂ after washing with water were also investigated by XPS analysis. Livraghi et al. [15] reported that the intensity of the N1s XPS peaks around 399–400 eV of nitrogen-doped TiO₂ prepared by a sol-gel method decreased and, in some cases, disappeared after washing with water in an ultrasonic bath. The values of P/Ti for P-TiO₂ after washing for 1 h (Table 2) are close to those of P/Ti before washing. The phosphorus atoms on the surface of P-TiO₂ were not eliminated by water.

3.5. Action spectra of acetaldehyde degradation

Action spectra were analyzed in order to investigate the wavelength range with visible-light photocatalytic activity. Fig. 5 shows the action spectra of photocatalytic reaction for acetaldehyde decomposition.



Assuming that the formation of a molecule of CO₂ requires five pairs of e⁻ and h⁺, the apparent quantum efficiency Φ_{app} was calculated as a ratio of the molar rate of the CO₂ generation to the flux of incident photons [38–40]. The apparent quantum efficiency was also calculated as a product of efficiencies of photoabsorption and electron–positive hole utilization [39]. Since the action spectra of P-TiO₂ corresponded well with the optical absorption spectra, the apparent quantum efficiency for acetaldehyde decomposition by use of P-TiO₂ was found to depend on the efficiency of photoabsorption. The apparent quantum efficiency of P-TiO₂ in Fig. 5 is ~0.66% in the 410–440 nm wavelength range, while the efficiency of ST01 falls to negligible values in the same wavelength range. These results show that the action spectra reveal the visible light photocatalytic activity of P-TiO₂. The apparent quantum efficiency of P-TiO₂ at the wavelengths longer than 420 nm appears to be low compared to its absorption spectra. In the absorption spectrum of P-TiO₂, broad (flat) absorption, without giving photocatalytic activity, might be overlapped to the absorption due to P-doping, which is responsible for activity. This led to apparent blue shift of an action spectrum from the absorption spectrum. It is also noted that the UV light activity of P-TiO₂ is larger than that of ST01. Previous studies on nitrogen-doped TiO₂ with visible-light activity showed that the photocatalytic efficiency of N-doped TiO₂ under UV light was less than that of pure TiO₂ [11,41]. The possible reason for the decrease in UV-light activity is that the N-doping causes formation of defect sites that lead to a loss of crystallinity [41] or that the N-doping causes an increase of carrier-recombination centers in TiO₂ [11].

3.6. Photocatalytic degradation of phenol

Fig. 6 shows the photodegradation of phenol (A) and the appearance of 1,4-benzoquinone (B) in the presence of P-TiO₂

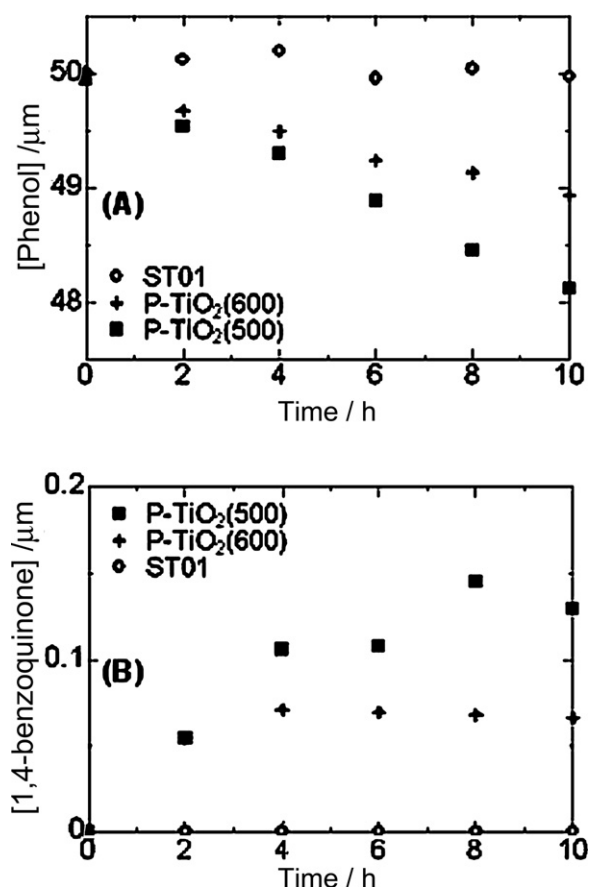


Fig. 6. Phenol degradation (A) and 1,4-benzoquinone formation (B) on P-TiO₂ and ST01 under visible light irradiation as a function of irradiation time.

and ST01 under visible light irradiation as a function of irradiation time. There was no degradation of phenol (A) or appearance of 1,4-benzoquinone (B) in the case of ST01 under visible light (423–600 nm) irradiation. On the other hand, the data for P-TiO₂ clearly show a decrease of phenol (A) and an increase of 1,4-benzoquinone (B). The experimental results indicate visible light activity of P-TiO₂ for phenol decomposition.

Fig. 7 shows apparent pseudo-first-order rate constants k_{app} for the photodegradation of phenol in the presence of P-TiO₂ as a function of calcination temperature. Up to a calcination temperature of 475 °C, the k_{app} value increases with increasing calcination temperature. On the other hand, the value above 475 °C decreases with increasing calcination temperature.

A large specific surface area causes a large amount of substrates to be adsorbed on TiO₂. Large specific surface areas are responsible

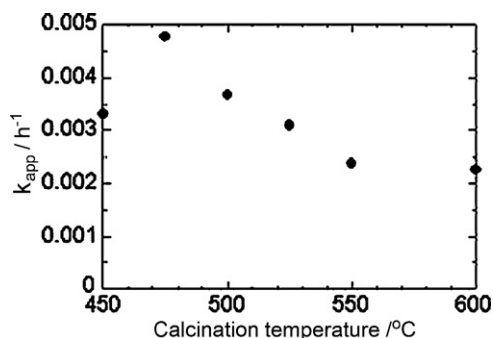


Fig. 7. Apparent rate constants k_{app} for photodegradation of phenol in the presence of P-TiO₂ as a function of calcination temperature.

for high photocatalytic efficiency [42]. In the present study, the specific surface area decreased with increasing calcination temperature as shown in Table 1. The large specific surface area in the range of 450–500 °C was associated with high photocatalytic efficiency at 475 °C.

3.7. Features of P-doped TiO₂

As for XPS analysis, a P2p peak of P-doped TiO₂ prepared using H₃PO₂ was observed around 132.6 eV [25]. In the present study, the peaks of P-TiO₂ were located around 134 eV (P⁵⁺) and 128 eV (P³⁻). The shape of absorbance for P-doped TiO₂ prepared using H₃PO₂ [26,27] was obviously different from that for P-TiO₂. The differences show that the properties of P-TiO₂ are not consistent with those of P-doped TiO₂ prepared using H₃PO₂. In the case of P/Ti in the range of 0.01–0.02, the absorption edges of P-TiO₂ and P-doped TiO₂ prepared using H₃PO₄ (P⁵⁺) [28] and H₃PO₂ (P³⁻) [26] were 422, 399 and 396 nm, respectively. The trend corresponds to the first-principles calculation [29]. The calculation indicated that the band gap of the doped TiO₂ had a slight narrowing for P anion-doped TiO₂ (anatase) which is prepared by replacement of lattice O²⁻ by P³⁻ and little narrowing for P cation-doped TiO₂ which is prepared by replacement of lattice Ti⁴⁺ by P⁵⁺.

The difficulty for doping TiO₂ with phosphorus having negative charges is that P species cannot be easily introduced into the TiO₂ lattice. Preparation by mixing phosphide and TiO₂ materials in solution provided definitive evidence of a red shift in the absorption edge for P-TiO₂. The dispersion of phosphide before calcination was found to be an important factor for the preparation of P-TiO₂.

During preparation of P-TiO₂, replacement of O²⁻ by P³⁻ produces oxygen vacancies. In order to maintain the overall electroneutrality of the crystal lattice, incorporation of two P atoms into the oxygen sites causes formation of one oxygen vacancy. Hydrogen-reduced TiO₂ powders were known as an example of having a large quantity of oxygen vacancies [7]. The powder exhibited a gray color, which indicates absorption over the entire range of visible light. Because the UV–vis spectra of P-TiO₂ show a continuous and tailing absorption below 500 nm, a small amount of oxygen vacancies in P-TiO₂ does not have much influence on the absorption. Incidentally, the oxygen vacancy state is not visible-light sensitive because hydrogen-reduced TiO₂ powders were not visible-light sensitive [7]. However, the oxygen vacancy is thought to act as a recombination center for holes and electrons because the oxygen vacancy state in anatase TiO₂ is below the lower end of conduction band at 0.75–1.18 eV [43]. Doping TiO₂ with an excess amount of P leads to a considerable decrease of photocatalytic activity for P-TiO₂. Further investigation to find an optimal doping amount of P is in progress.

4. Conclusions

Doping anatase TiO₂ with phosphorus having negative charges results in the preparation of P-doped TiO₂ with a definite red shift of the absorption edge and efficient visible-light photocatalytic activity. The dispersion of phosphide before calcination was found to be an important factor for the preparation of P-TiO₂. The absorption edge of P-TiO₂ was obviously larger than that of P-doped TiO₂ prepared using H₃PO₄ (P⁵⁺) and H₃PO₂ (P³⁻). The results are in agreement with the first-principles calculation for P anion-doped TiO₂. The action spectra of photocatalytic reaction for acetaldehyde decomposition proved the visible-light photocatalytic activity of P-TiO₂. The present study indicates that P³⁻-doped TiO₂ should be investigated as a new material of nonmetal-doped TiO₂ that can yield clear reactivity under visible light.

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